Prototype Structure of Pb₅Ge₃O₁₁*

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The prototypic structure of the optically active ferroelectric $Pb_5Ge_3O_{11}$ has been deduced from packing considerations and refined from single-crystal diffractometer data. Interleaving segments of the apatite and nasonite structures in layers perpendicular to c gives the lead germanate structure. The space group of the prototype is $P\delta$, with a = 10.190, c = 10.624 Å, and Z = 3. Atomic coordinates were refined to give an *R*-factor of 0.093 using 463 X-ray reflections. The structure contains tetrahedral GeO₄ groups and double tetrahedral Ge₂O₇ groups in equal numbers. Lead is found in both trigonal pyramid and trigonal prism configurations.

Lead germanate $(Pb_5Ge_3O_{11})$ has attracted considerable interest because it is both ferroelectric (1, 2) and optically active (3, 4). Below the Curie temperature of 177°C, the optical handedness inverts when the spontaneous polarization is reversed by an applied electric field. Unlike most ferroelectrics, 180° domains are clearly visible because of the optical rotatory power of 5.6°/mm for 5890 Å light. Using a modification of the Czochralski growth technique developed in Japan (5), we have grown clear transparent crystals exhibiting a well-defined domain structure (6). With limited-area switching, the domain states may prove useful for information storage and in the processing of light signals. Additional work on Pb₅Ge₃O₁₁ include studies of the piezoelectric and elastic properties (7), and the Raman spectrum (8-10) which shows an overdamped soft mode near the transition temperature. Based on physical properties, the Shuvalov symbol for the transition in lead germanate is $\overline{6}(1)D\overline{6}F3$.

Experimental Methods

A sphere of $Pb_5Ge_3O_{11}$ with a radius of 35 μ m was ground from a fragment of a Czochralski-

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grown crystal using a compressed air grinder. Least-squares refinement of the orientation matrix yielded lattice parameters of a = 10.190(6)and c = 10.624(5) Å. The unit cell contains 3 formula units in a volume 955.3 Å³, giving a calculated specific gravity of 7.41, slightly larger than the measured value of 7.33(5). Intensity data for 463 reflections were collected using an automated diffractometer with a 2θ scan and filtered Mo $K\alpha$ radiation. Since there were no systematic absences, or superlattice reflections, the space group of the prototype structure is $P\delta$ and that of the ferroelectric phase is P3. Standard reflections monitored periodically throughout data collection showed an extreme spread of 6%and a standard deviation from the mean of 1%. Intensities were corrected for absorption using the spherical correction factors given in Vol. 2 of the International Tables for X-Ray Crystallography. The maximum absorption correction (30.9) was less than three times the minimum (11.3).

A trial structure for $Pb_5Ge_3O_{11}$ was deduced from the apatite (11) and nasonite (12) structures. Cell constants and powder patterns for the three materials show a strong resemblance. The *a* dimension of all three cells is about 10 Å, while *c* is approximately 7 Å for apatite, 10 Å for lead germanate and 13 Å for nasonite. The apatite structure contains two layers of isolated tetrahedra related by a 6_3 screw axis parallel to *c*.

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FIG. 1. Comparison of the idealized apatite, nasonite, and lead germanate structures showing a section parallel to (010) about 4 Å thick.

Nasonite has the same symmetry but contains double tetrahedra instead of discrete tetrahedra. Sections of the two structures are illustrated in Fig. 1. Combining half the apatite unit cell with half the nasonite cell gives the lead germanate structure, as shown in Fig. 1c. The trial model belonging to space group $P\delta$ was confirmed by subsequent refinement.

The Bell-ORFLS (13) least-squares program was used for the structure refinement. There are 7 Pb, 5 Ge and 18 O coordinates to be refined in the $P\delta$ prototype structure. Adding isotropic temperature factors for the 15 atoms in the asymmetric unit gives a total of 45 variable parameters in the least-squares refinement.

Beginning with the trial prototype structures derived from packing considerations, the Pb atomic coordinates were varied first, then the Ge coordinates, followed by Pb and Ge isotropic temperature factors. Next, all Pb and Ge parameters were varied simultaneously. The same procedure was used for oxygen coordinates and temperature factors. A least squares refinement varying all 45 parameters yielded an *R*-factor of 0.127. Isotropic temperature factors were converted to anisotropic form giving a total of 94 independent parameters. Two least-squares cycles gave an R-factor of 0.10. Two final cycles omitting 45 reflections below the observable limit yielded an R-factor of 0.093. Structure factors computed with the final coordinates are listed in Table I. There were no correlations involving atomic positions greater than 0.5. Anisotropic temperature factors for two of the atoms were not positive-definite but the discrepancies were within standard error. It is possible that the negative temperature factors reflect a systematic error such as absorption.

Since the orientation of a was arbitrarily assumed when the intensities were measured, the h and k indices of the reflections were interchanged to test the second possible orientation. Refined coordinates giving an agreement index of 0.093 for hkl reflections gave 0.136 for khlreflections after two least-squares cycles. Moreover, the new coordinates led to unacceptable interatomic distances, confirming the original choice of axes.

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Asterisks denote reflections omitted from least-squares refinement.

Pb₅Ge₃O₁₁ STRUCTURE

TABLE I

Atom	Equipoint	x	у	Z	B (Å ²)
Pb1	61	0.2622(5)	0.2628(5)	0.1849(4)	0.8(1)
Pb ₂	2h	0.333	0.667	0.3382	-0.2(2)
Pb ₃	2 <i>i</i>	0.667	0.333	0.325(1)	2.4(3)
Pb₄	1 <i>c</i>	0.333	0.667	0.000	1.0(2)
Pb₅	1 <i>e</i>	0.667	0.333	0.000	2.2(3)
Pb ₆	3 <i>k</i>	0.249(1)	0.989(1)	0.500	2.1(2)
Ge1	61	0.015(1)	0.388(1)	0.152(1)	0.9(3)
Ge ₂	3 <i>k</i>	0.395(2)	0.389(2)	0.500	0.4(3)
O 1	61	0.074(11)	0.330(8)	0.268	3.7(1.8)
02	61	0.074(8)	0.581(9)	0.155(8)	3.7(1.5)
O3	61	0.818(16)	0.329(11)	0.195(11)	6.0(3.0)
O4	3 <i>j</i>	0.066(6)	0.347(7)	0.000	0.3(1.2)
O5	3k	0.286(7)	0.481(8)	0.500	-0.8(1.5)
O ₆	3 <i>k</i>	0.584(10)	0.453(13)	0.500	2.9(2.1)
07	61	0.302(16)	0.242(14)	0.377(8)	5.4(3.0)

Equipoints, Atomic Coordinates, and Isotropic Temperature Factors for the Lead Germanate Structure Refined in Space Group $P\delta^a$

" Standard errors associated with the final digit are given in parentheses.

Discussion

Atomic coordinates and isotropic temperature factors for Pb₅Ge₃O₁₁ are presented in Table II. The relatively high standard errors and large temperature factors no doubt arise partly because the refinement was carried out with low angle data in the prototype space group $P\bar{6}$, rather than using the true space group P3 and a more complete data set. Nevertheless, interatomic distances calculated with these coordinates indicate that the structure is satisfactory for the prototype model. Metal-oxygen distances are reasonable and the shortest O-O contact in the structure is 2.62 Å.

Table III lists the coordination number and average bond lengths for the metal atoms. Germanium atoms are in tetrahedral coordination with an average Ge-O distance of about 1.77 Å, in good agreement with the value 1.74 Å obtained from several well-refined structures containing GeO₄ tetrahedra (14). Lead atoms are found in threefold triangular pyramids and sixfold triangular prisms. Average Pb-O distances range from 2.23 to 2.86 Å. For comparison, in red PbO, lead forms tetragonal pyramids with an average interatomic distance of 2.30 Å (15), while in PbTiO₃ the average Pb-O distance is 2.81 Å (16).

Figures 2 and 3 show (001) sections of the GeO₄ and Ge₂O₇ layers corresponding to the apatite and nasonite segments of the structure. Open channels, similar to those found in several lead apatites, lie along the c axes.

We are presently collecting neutron diffraction data and high-angle X-ray intensities to refine the

TABLE III

Atom	Oxygen coord. no.	Coord. polyhedron	Av bondlength (Å)
Pb ₁	3	Triangular pyramid	2.23
Pb ₂	3	Triangular pyramid	2.41
Pb3	6	Triangular prism	2.33
Pb ₄	6	Triangular prism	2.86
₽b₅	6	Triangular prism	2.60
Pb_6	6	Pentagonal pyramid	2.61
Get	4	Tetrahedron	1.75
Ge₂	4	Tetrahedron	1.79



FIG. 2. Section of $Pb_5Ge_3O_{11}$ projected slightly off (001). The GeO₄ layer and interlayer lead coordination polyhedra are shown.

structure of the ferroelectric phase. Although it is premature to speculate on the atomic motions responsible for the spontaneous polarization and optical activity, one possible mechanism appears obvious. The lead atoms located in triangular prisms may shift along c, changing to triangular pyramid coordination. Such a motion would polarize the crystal along c, destroying the mirror plane perpendicular to c and lowering the symmetry from $\overline{6}(=3/m)$ to 3. Optical activity could result from accompanying motions of the neighboring oxygens. A shift in the Pb position along +c would impart a clockwise twist to the germanium tetrahedra, giving rise to a handedness similar to quartz. A downward shift creates a counterclockwise twist, so that polarization reversal is accompanied by a change in handedness, and a change in sign of the optical activity coefficient.

Acknowledgments

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FIG. 3. Section of Pb₅Ge₃O₁₁ projected slightly off (001), showing the Ge₂O₇ nasonite-like layer.

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